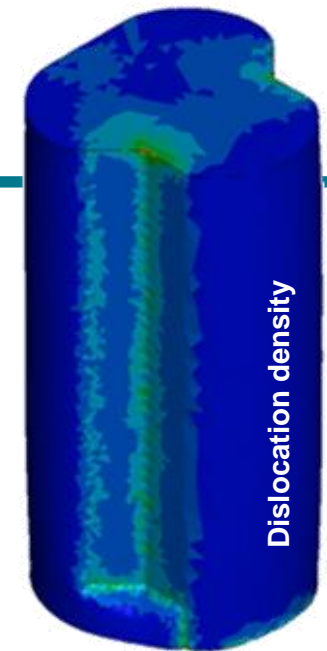


On the prediction of process-dependent material properties for Ti-6Al-4V with LS-DYNA

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Agenda

- Motivation
- Phase change algorithms in *MAT_254 for Ti-6Al-4V
- New generalized yield stress definition in *MAT_254 in LS-DYNA
- Numerical examples
- Summary and Outlook

Motivation

Innovative manufacturing processes for Ti-6Al-4V

- Combination of different state-of-the-art manufacturing processes

- Additive manufacturing and hotforming

- Forging and additive manufacturing

- Replacing or adding process steps

- Can enhance the process efficiency

- Affects the microstructure of the material

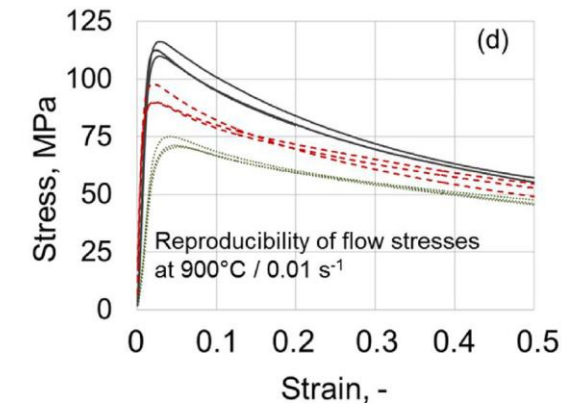
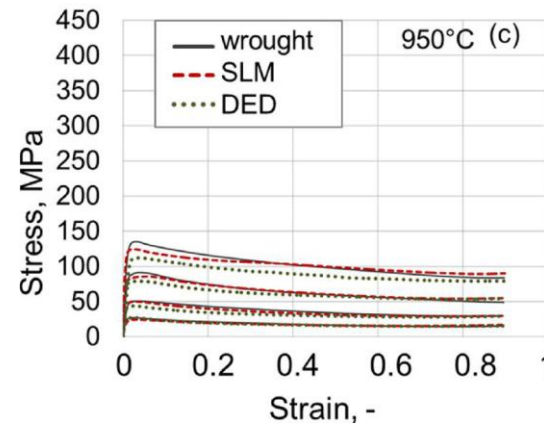
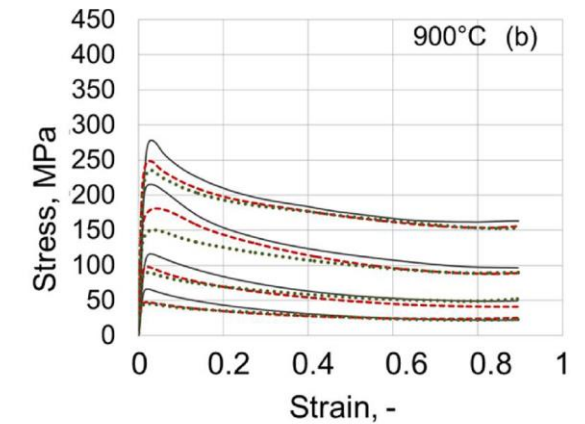
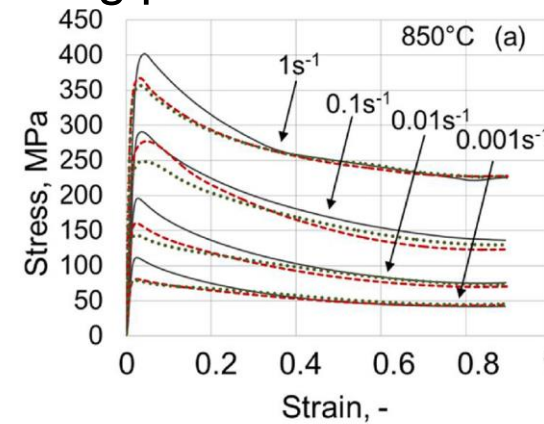
- Leads to different mechanical behavior

- Goal:

Predict the resulting mechanical properties by numerical simulation with LS-DYNA

Literature:

Bambach M, Sizova I, Szyndler J, Bennett J, Hyatt G, Cao J, Papke T, Merklein M. On the hot deformation behavior of Ti-6Al-4V made by additive manufacturing. Journal of Materials Processing Technology, Volume 288, 2021, 116840

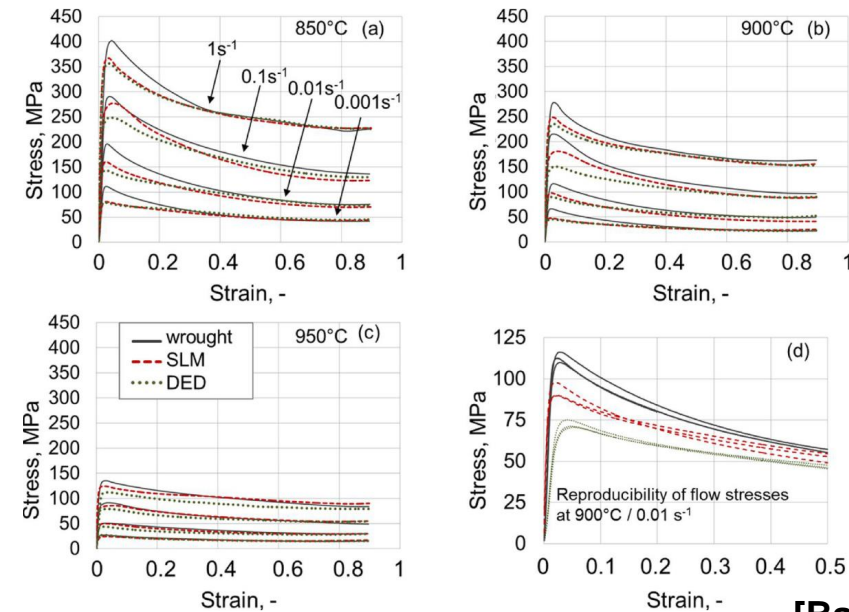


Motivation

Necessary steps towards this goal

- Capture phase changes in Ti-6Al-4V
 - Use material model *MAT_254 due to
 - Powerful phase transformation framework
 - Flexible tabular input structure
 - See [Kloppel et.al, Esaform 2020]
- Predict yield stress
 - as complex function of
 - Strain, strain rate, temperature
 - Microstructure properties such as phase distribution, dislocation density, globularization, ...
 - See [Buhl et.al, Esaform 2021]

	to phase				
	1	2	3	...	n
1		data ₁₂	data ₁₃	...	data _{1n}
2	data ₂₁		data ₂₃	...	data _{2n}
3	data ₃₁	data ₃₂		...	data _{3n}
...	⋮	⋮	⋮		⋮
n	data _{n1}	data _{n2}	data _{n3}	...	



[Bambach et al., 2021]

Phase change algorithms

Implementations in *MAT_254 for Ti-6Al-4V

Phase change algorithms

Phase transformation framework in *MAT_254

- Up to 24 individual phases (= 552 possible phase change scenarios)
- Phase changes in heating, cooling or in a temperature window
- User can choose from a list of phase change models for each scenario

	1	2	3	4	5	6	7	8
Card 3	PTLAW	PTSTR	PTEND	PTX1	PTX2	PTX3	PTX4	PTX5
Card 4	PTTAB1	PTTAB2	PTTAB3	PTTAB4	PTTAB5	PTTAB6	PTTAB7	PTTAB8

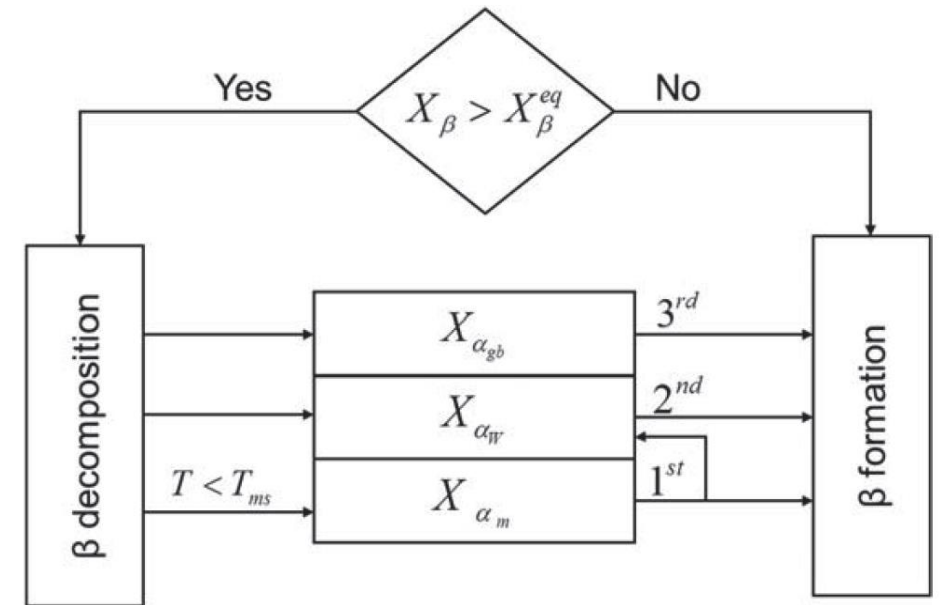
- Parametrization to be given in a tabular structure, i.e. matrix input (*DEFINE_TABLE_2D/3D) for
 - Phase transformation law (2D)
(Koistinen-Marburger, JMAK, Åkerström, Oddy, ...)
 - Start and end temperatures, temperature window (2D)
 - Transformation constants (2D)
 - Temperature (rate) dependent parameters (3D)

		to phase				
		1	2	3	...	n
from phase	1		data ₁₂	data ₁₃	...	data _{1n}
	2	data ₂₁		data ₂₃	...	data _{2n}
	3	data ₃₁	data ₃₂		...	data _{3n}
	...	⋮	⋮	⋮		⋮
	n	data _{n1}	data _{n2}	data _{n3}	...	

Phase change algorithms

Microstructure Model of Ti-6Al-4V

- Based on [Charles Murgau et al., 2012] and [Klusemann and Bambach, 2018]
- Considered phases
 - β -phase, stable at higher temperatures
 - Variants of α -phases, stable at lower temperatures:
 - Grain boundary phase α^{gb}
 - Widmanstätten phase α^w
 - Martensitic/massive phase α^m
- Phase transformation based on equilibrium X_β^{eq}



[Charles Murgau, PhD-thesis, 2016]

Literature:

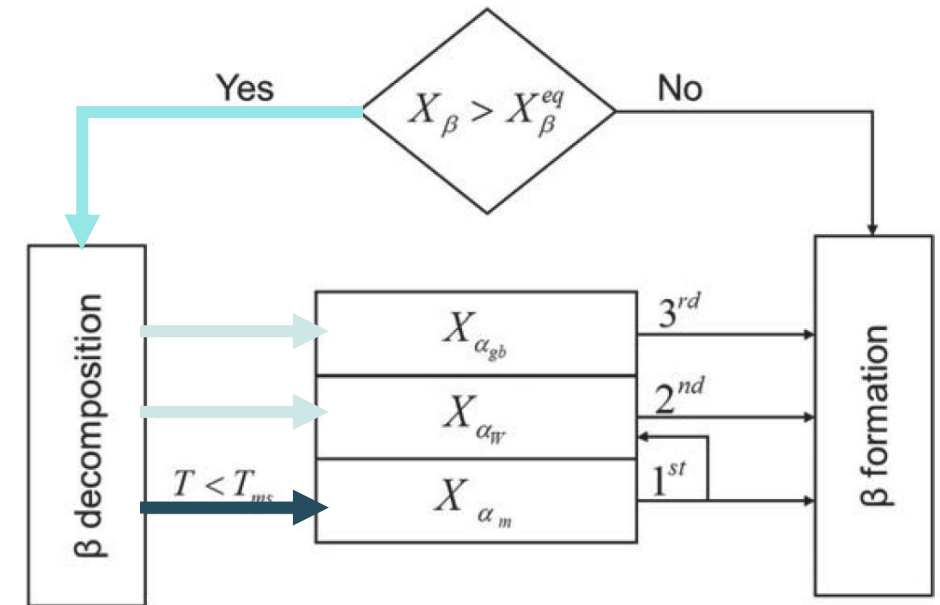
Charles Murgau, C.; Pederson, R.; Lindgren, L. E.; "A model for Ti-6Al-4V microstructure evolution for arbitrary temperature changes"; Modelling Simul. Mater. Sci. Eng. 20 (2012) 055006 (23pp)

Klusemann, B.; Bambach, M.; "Stability of phase transformation models for Ti-6Al-4V under cyclic thermal loading imposed during laser metal deposition"; AIP Conference Proceedings 1960 140012 (2018)

Phase change algorithms

*MAT_254 for Ti-6Al-4V in cooling

- Diffusion-controlled mechanism for formation of α^{gb} and α^w
 - Straight-forward extension of JMAK model (**law 2**)
 - Transformation only active if equilibrium $x_{eq,a}$ of source is exceeded
- Diffusionless transformation of β into α^m
 - As generalized Koistinen-Marburger model (new **law 9**)
 - Standard Koistinen-Marburger for a rapid cooling
 - Incomplete Koistinen-Marburger for intermediate cooling rates, where some of the source phase remains
 - No transformation for slow cooling
 - Transformation only active if equilibrium $x_{eq,a}$ of source is exceeded

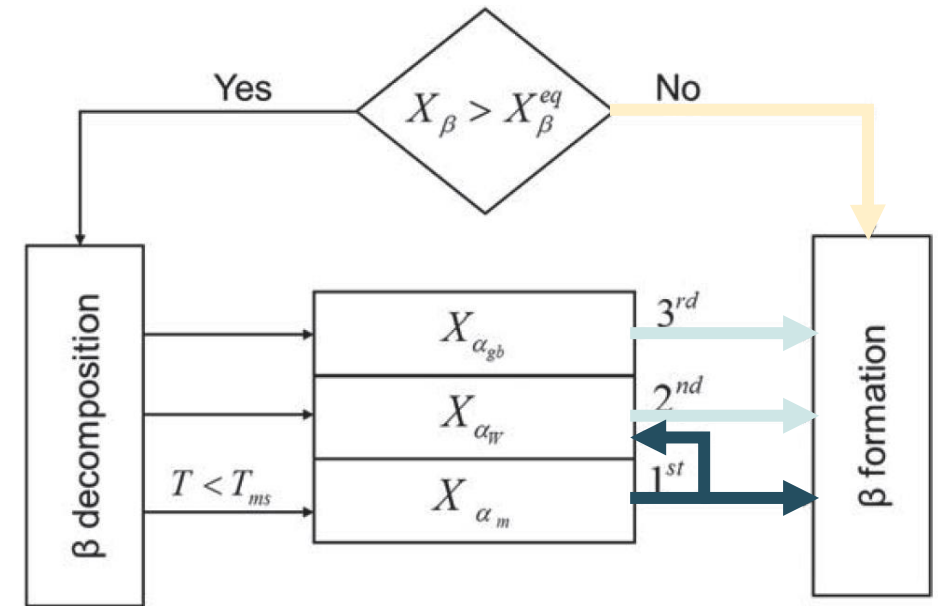


[Charles Murgau, PhD-thesis, 2016]

Phase change algorithms

*MAT_254 for Ti-6Al-4V in heating

- Sequential, diffusion of α^w and α^{gb} into β
 - **Law 7** (parabolic growth, for $\alpha^w \rightarrow \beta$)
 - Model determines remaining fraction of a group of phases
 - Source phase is the first one to be dissolved
 - **Law 8** (sequential dissolution, for $\alpha^{gb} \rightarrow \beta$)
 - Dissolve further members of the group in a given order
- Simultaneous decomposition of α^m into α^w and β
 - **Law 5** for $\alpha^m \rightarrow \beta$
 - Dissolution of source phase into a virtual (“recovered”) phase
 - Partial transformation of virtual phase into target phase
 - **Law 6** for $\alpha^m \rightarrow \alpha^w$
 - Partial transformation of virtual phase as calculated by previous transformation into target phase



[Charles Murgau, PhD-thesis, 2016]

Yield stress definition

New generalized yield stress definition in *MAT_254

Yield Stress Definition

Novel interface for the yield stress input in *MAT_254

Description for β -phase according to literature

$$\sigma_{\beta} = \sigma_{1\beta} - \sigma_{2\beta} \quad [\text{Bambach et al., 2021}]$$

$$\sigma_{1\beta} = \begin{cases} \sigma_p \left(\frac{\varepsilon}{\varepsilon_p} \exp\left(1 - \frac{\varepsilon}{\varepsilon_p}\right) \right)^c & \text{for } \varepsilon < \varepsilon_p \\ \sigma_p & \text{for } \varepsilon \geq \varepsilon_p \end{cases}$$
$$\sigma_{2\beta} = \begin{cases} 0 & \text{for } \varepsilon < \varepsilon_{p\beta} \\ (\sigma_p - \sigma_s) \left(1 - \exp\left(-k \left(\frac{\varepsilon - \varepsilon_{p\beta}}{\varepsilon_{ss} - \varepsilon_{p\beta}}\right)^n\right) \right) & \text{for } \varepsilon \geq \varepsilon_{p\beta} \end{cases}$$

$$\varepsilon_{p\beta} = \varepsilon_p / \beta, \quad \varepsilon_{ss} = E_1 Z^{E_3}, \quad \varepsilon_p = A_1 Z^{A_3},$$

$$\sigma_p = D_1 \operatorname{asinh}\left(\left(\frac{Z}{D_2}\right)^{D_3}\right), \quad \sigma_s = C_1 \operatorname{asinh}\left(\left(\frac{Z}{C_2}\right)^{C_3}\right)$$

$$Z = \dot{\varepsilon} \exp\left(\frac{Q_w}{R T}\right)$$

Material input for yield stress of β -phase

- Use *DEFINE_FUNCTION interface
 - Needs access to strain, strain rate and temperature

```
*DEFINE_FUNCTION
5001
float yieldbeta(float time, float T, float epspl,
float epsplr, float epseqv, float epseqvrt)
{...
Z= epseqvrt*EXP(Q/(R*T));
sigp=C1*ASINH((Z/C2)**C3);
sigs=D1*ASINH((Z/D2)**D3);
epsp=A1*(Z**A3);
epss=E1*(Z**E3);
...
sig1=sigp*(epseqv/epsp*EXP(1-(epseqv/epsp))**C;
sig2=(sigp-sigs)*
(1-EXP(-k*((epseqv -epsp/beta)/(epss-epsp/beta))**n));
...
sig=sig1-sig2;
return sig;}
```

Yield Stress Definition

Novel interface for the yield stress input in *MAT_254

Description for α -phase according to literature

$$\sigma_{\alpha} = M[\tau^* + \tau_{\mu}] \quad [\text{Bambach et al., 2021}]$$

$$\tau^* = \tau^0 \left(1 - \left(\frac{RT}{\Delta G} \ln \frac{\dot{\epsilon}_{\text{ref}}}{\dot{\epsilon}} \right)^{1/q} \right)^{1/p}$$

$$\tau_{\mu} = \tau_0 + \alpha_2 G b \sqrt{\rho} + K_{HP} L^{-1/2}$$



Interface data for α -phase in MAT_254

- Function needs access to
 - strain, strain rate and temperature
 - internal variables and shear modulus G

```
*DEFINE_FUNCTION
    5002
float yieldalpha(float time, float T, float epspl,
float epsplr, float epseqv, float epseqvrt,
float shrm, float bulk, float intvar1, float intvar2,
float intvar3, float intvar4)
{
...

taust=tau0*(1-
(R*T/deltaG*log(epsrefrt/epseqvrt))**(1/q))**(1/p);
taumu=tau0+alpha2*shrm*b*sqrt(intvar2)+L**(-0.5)*intvar3;

sig=M*(taust+taumu);
return sig;
}
```

- Dislocation density ρ and Hall-Petch coefficient K_{HP}
 - Cannot be given in a closed form
 - Evolution follows an ODE
 - Are incorporated as internal variables in the material formulation

Yield Stress Definition

Concept of internal variables - input

	1	2	3	4	5	6	7	8
Card 1	MID	RHO	N	E	PR	MIX	MIXR	
Card 2	TASTART	TAEND	TABCTE				DTEMP	NIVAR
Card 2a	XASTR	XAEND	XAPAR1	XAPAR2	XAPAR3		CTEANN	
Card 5	PTEPS	TRIP	PTLAT	POSTV	NUSHIS	GRAI	T1PHAS	T2PHAS
Card 5a	FUNUSH1	FUNUSH2	FUNUSH3	FUNUSH4	FUNUSH5	FUNUSH6	FUNUSH7	FUNUSH8
IVAR=1, 1st add-on	IVAROPT	INITVAL	IVARPAR1	IVARPAR2	IVARPAR3	IVARPAR4	IVARPAR5	IVARPAR6
IVAR=1, 2nd add-on	IVARPAR7	IVARPAR8	IVARPAR9	IVARPAR10	IVARPAR11	IVARPAR12	IVARPAR13	IVARPAR14
⋮	⋮	⋮	⋮	⋮	⋮	⋮	⋮	⋮
IVAR=NIVAR, 1st add-on	IVAROPT	INITVAL	IVARPAR1	IVARPAR2	IVARPAR3	IVARPAR4	IVARPAR5	IVARPAR6
IVAR= NIVAR, 2nd add-on	IVARPAR7	IVARPAR8	IVARPAR9	IVARPAR10	IVARPAR11	IVARPAR12	IVARPAR13	IVARPAR14
Card 6	LCY1	LCY2	LCY3	LCY4	LCY5	LCY6	LCY7	LCY8

Yield Stress Definition

Concept of internal variables - evolution

- For Ti-6Al-4V, three different types of internal variables are needed:
 - percentage X of globular α
 - dislocation density ρ
 - Hall-Petch coefficient K_{HP}

- Evolution equations of X and ρ depend on each other

■ IVAROPT

- = 3: calculate percentage X of globular α
- = 4: calculate dislocation density ρ
- = 5: refers to the Hall-Petch coefficient K_{HP}

$$\dot{X} = \kappa_7(1 - X)\dot{\varepsilon}^{\kappa_8}(\varepsilon - \varepsilon_C)^{\kappa_9} \frac{\kappa_{10} \rho G}{T} \exp\left(\frac{\kappa_{11}}{T}\right)$$

$$\varepsilon_C = \kappa_1 \dot{\varepsilon}^{\kappa_2} \left(\exp\left(\frac{\kappa_4}{T}\right)\right)^{\kappa_3}$$

$$\dot{\rho} = \begin{cases} (\kappa_7\sqrt{\rho} - k_2\rho)\dot{\varepsilon} & , \quad \text{for } \varepsilon < \varepsilon_C \\ (\kappa_7\sqrt{\rho} - k_2\rho)\dot{\varepsilon} - \frac{\dot{X}\rho(\varepsilon - \varepsilon_C)}{1 - X} & , \quad \text{for } \varepsilon > \varepsilon_C \end{cases}$$

$$\varepsilon_C = \kappa_1 \dot{\varepsilon}^{\kappa_2} \left(\exp\left(\frac{\kappa_4}{T}\right)\right)^{\kappa_3} , \quad k_2 = \kappa_8 \left[\dot{\varepsilon} \left(\exp\left(\frac{\kappa_9}{T}\right)\right)^{\kappa_{10}}\right]^{\frac{1}{\kappa_{11}}}$$

$$\dot{\lambda} = \kappa_1(\kappa_2 - \lambda)\dot{\varepsilon}$$

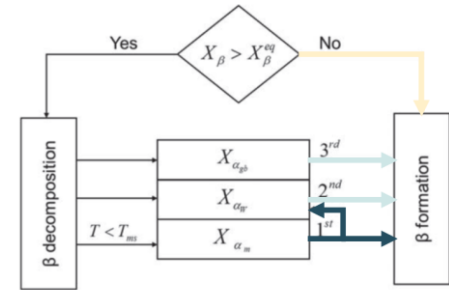
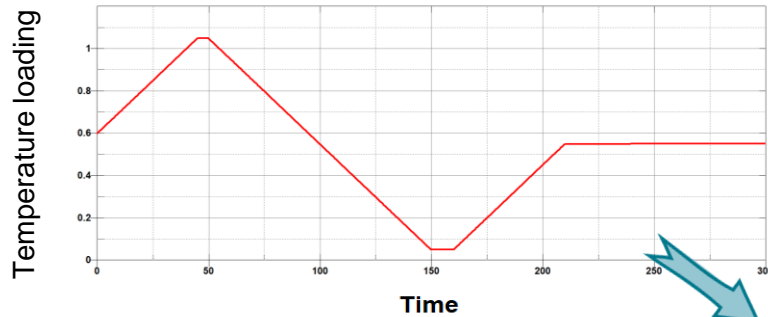
$$K_{HP} = \lambda \sqrt{G\kappa_3} \kappa_4 (\dot{\varepsilon})^{\kappa_5} \left(\exp\left(\frac{\kappa_7}{T}\right)\right)^{\kappa_6} f_\alpha^{\kappa_8}$$

[M. Bambach et al., 2021]

Numerical Examples

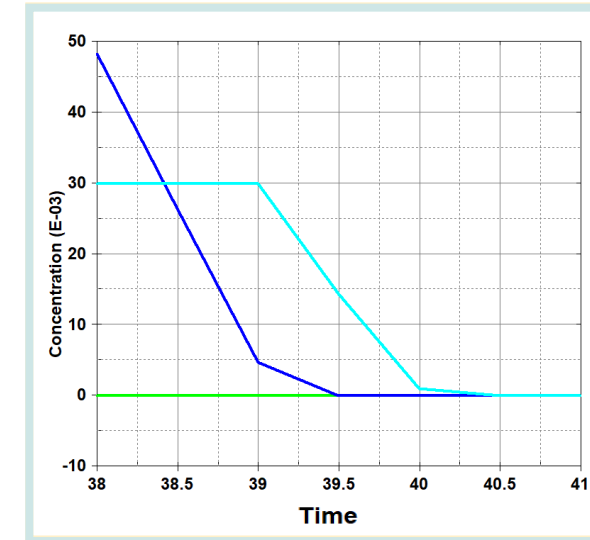
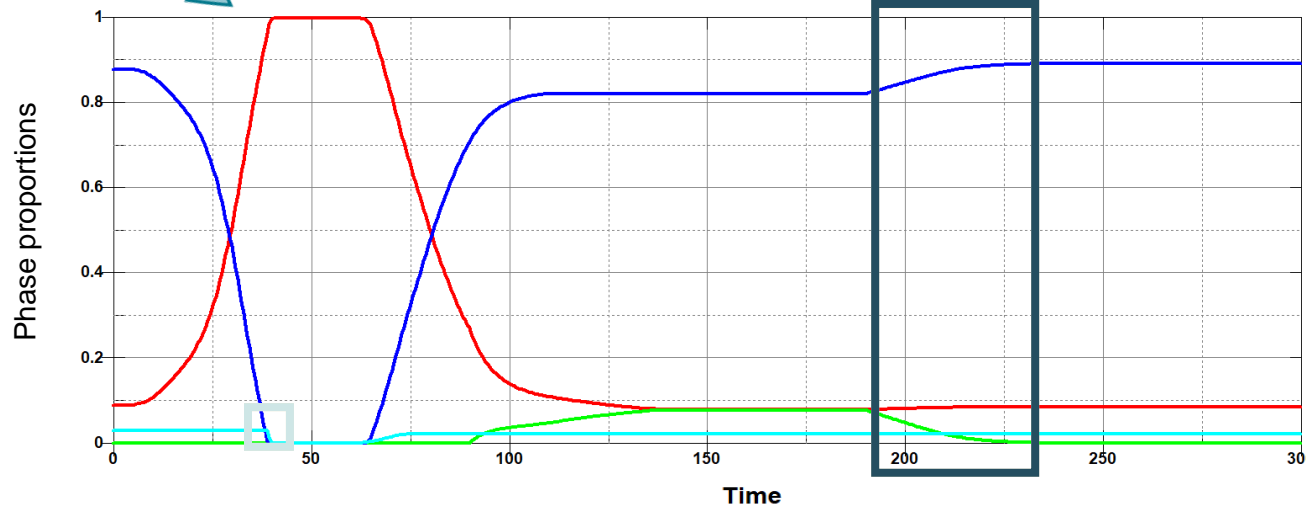
Numerical Examples

Single element model with prescribed temperature loading



[Charles Murgau, PhD-thesis, 2016]

- A beta phase
- B martensitic alpha
- C Widmannstätten alpha
- D grain boundary alpha

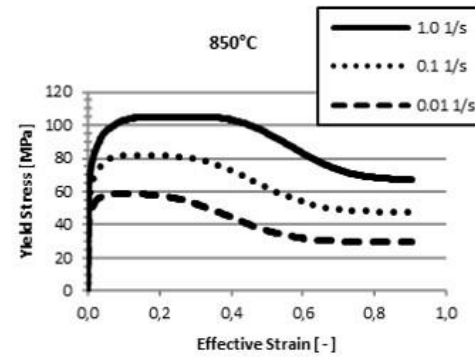


- Simultaneous decomposition of α^m into α^w and β
- Sequential decomposition of α^w and α^{gb} into β

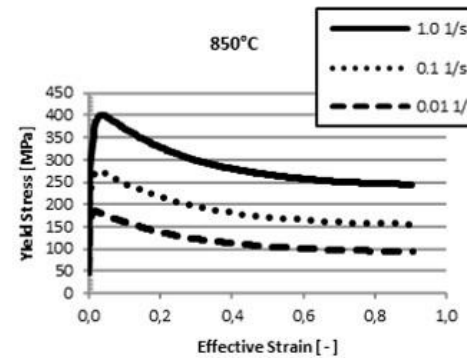
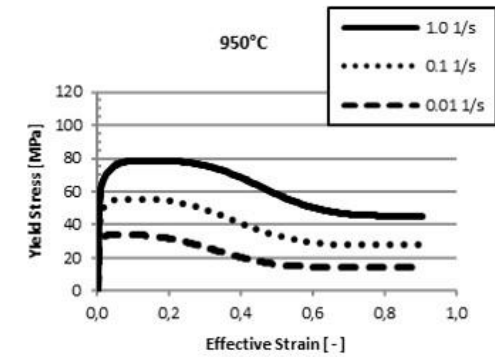
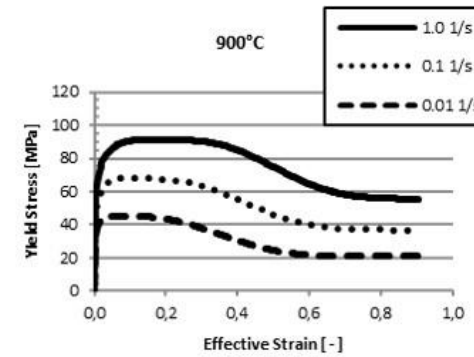
Numerical Examples

Yield stress for single element models I

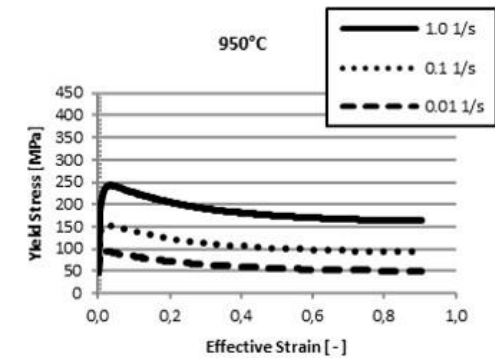
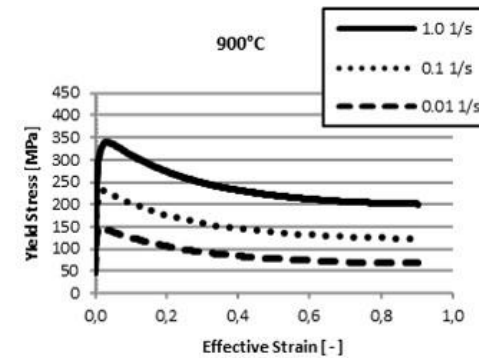
- Constant temperatures
- Uni-axial loading with constant velocities
- No phase transformations active
- Test β - and α - phases individually



β - phase



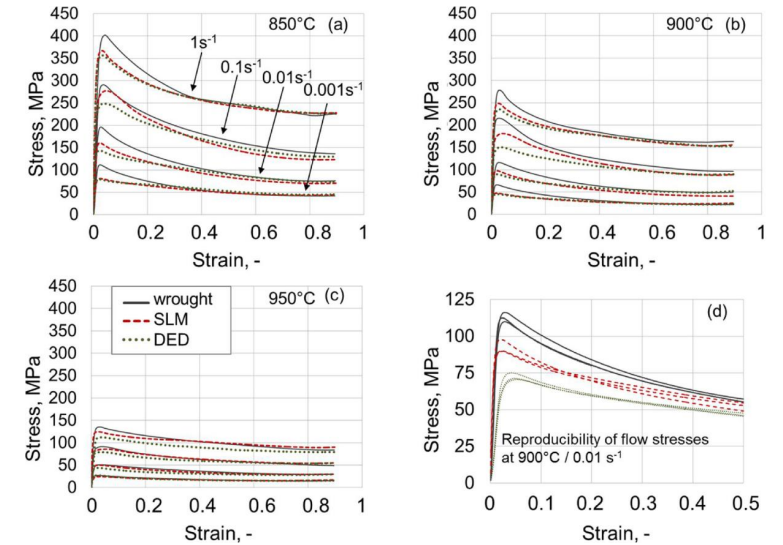
α - phase



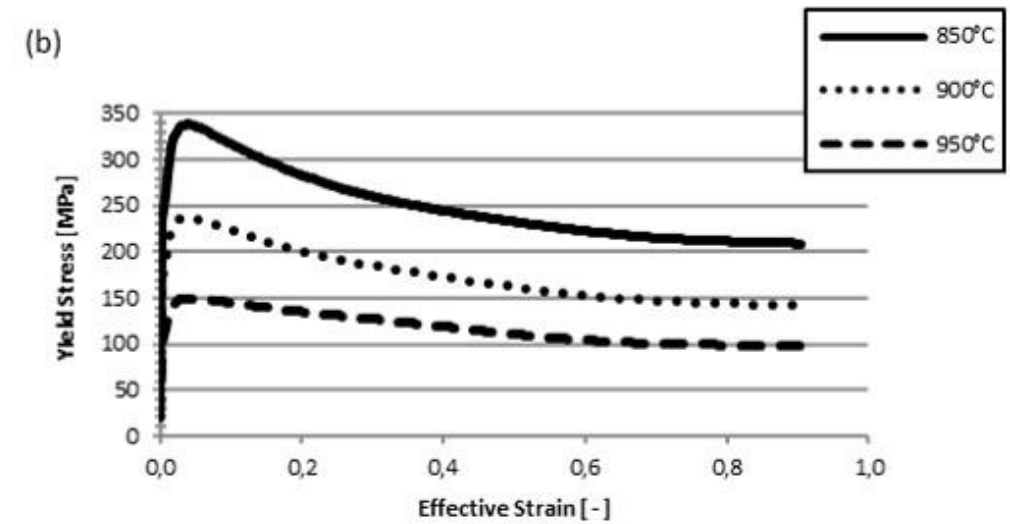
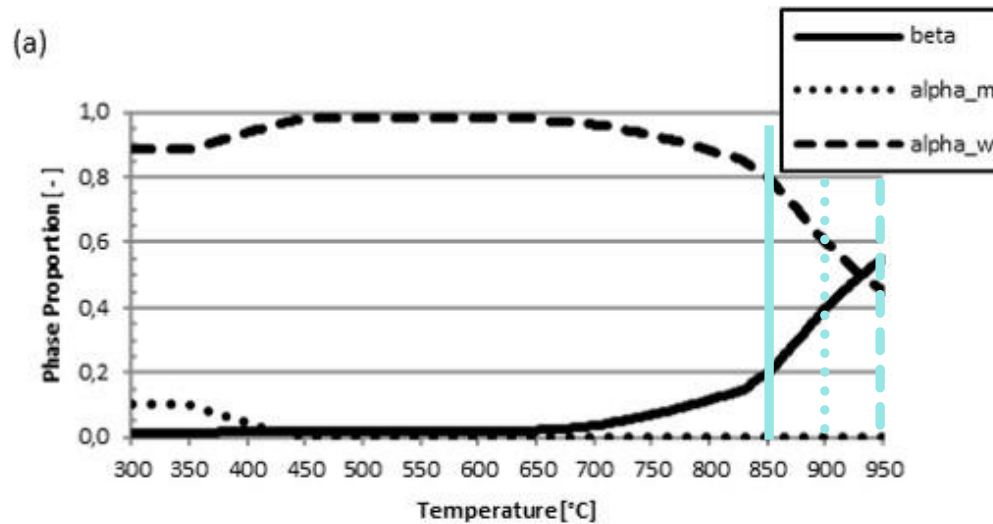
Numerical Examples

Yield stress for single element models II

- Active phase change algorithm
- Increase temperature linearly
- Slow temperature rate to guarantee equilibrium
- Uni-axial loading with constant strain rate 1/s



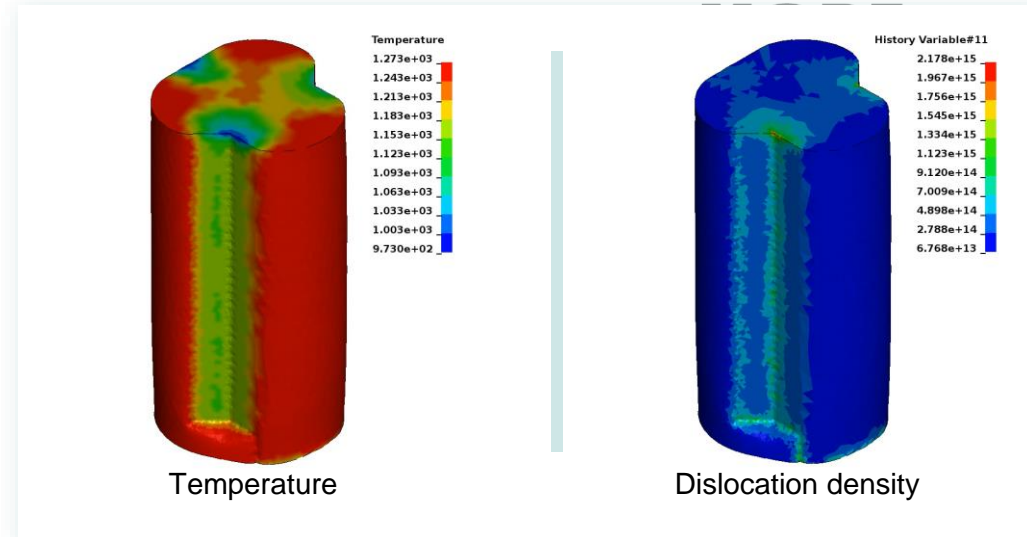
[Bambach et al., 2021]



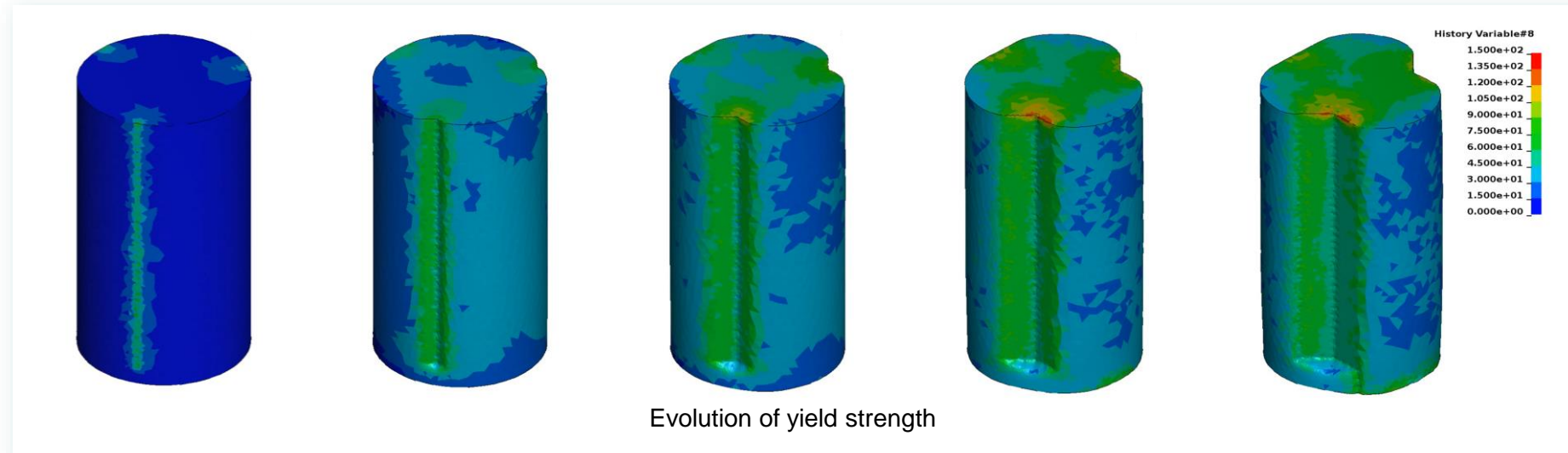
Numerical Examples

Reduced forging simulation

- Forging of a hot cylinder
 - Initial temperature cylinder: 1000°C
 - Tool temperature: 27°C
 - EFG-approach with remeshing



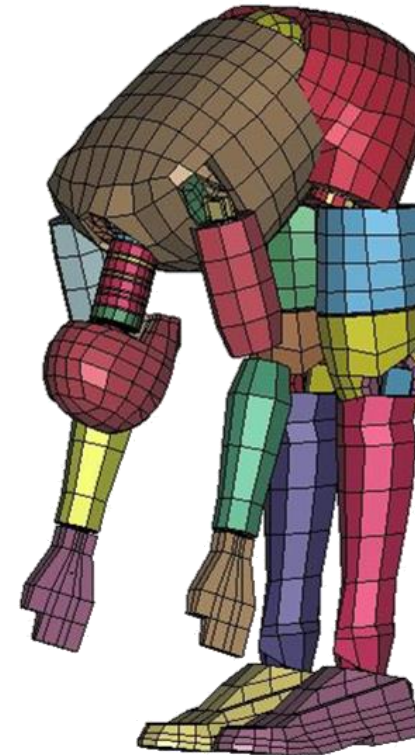
- Plausible evolution of
 - Temperature
 - Dislocation density
 - Yield strength



Summary and Outlook

- Discussed phase change algorithms for Ti-6Al-4V in *MAT_254
 - Easily incorporated in the framework
 - Novel concepts particularly for beta formation (dissolution of groups, recovery of martensitic alpha)
- Introduced a new concept of internal variables in *MAT_254
 - Captures certain evolution effects on the microstructure (dislocation density, globularization, Hall-Petch coefficient)
 - Concept not restricted to the presented effects
- Presented a general interface for definition of yield stress
 - Based on *DEFINE_FUNCTION keyword
 - Accesses a variety of strain and stress data and internal variables
- With these enhancements of *MAT_254 we predict the flow curves of Ti-6Al-4V
- Apply concepts to other classes of material and to other applications

Thank You



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